Copolymers of Ethylene with Butene-1 And Long Chain a-Olefins. II. Dodecene-1 as Long Chain a-Olefin.

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Synopsis

Ethylene/dodecene-1 and ethylene/butene-1/dodecene-1 copolymers were prepared and their polymerization reaction in heptane slurry was studied. The Ziegler-Natta catalyst was aluminium-activated $TiCl_3$, and the cocatalyst was triethylaluminium. The dodecene-1 contents in the ethylene/dodecene-1 copolymers were less than 3 wt %. The dodecene-1 contents in terpolymers of ethylene, butene-1, and dodecene-1 ranged up to 10 wt %. The dependence of basic properties on composition of terpolymers was investigated.

INTRODUCTION

Properties of polyethylene are dependent on branch length and distribution. To produce side chains of polyethylene in suspension or gas phase processes, usually less than 6 carbon atoms containing α -olefins are used. In solution processes longer α -olefins may be used.

In Part I,¹ the reaction of decene-1 with ethylene in suspension was discussed. The reaction rate of decene-1 increased, when butene-1 was added simultaneously. In the terpolymers obtained, decene-1 content ranged up to 13.5 wt % when density was 914 g/dm³. The reaction of dodecene-1 in the corresponding system was studied. Investigations of the mechanism and comonomer distribution were also performed.

EXPERIMENTAL

Polymerizations with dodecene-1 were carried out using the same procedures as reported in Part I.¹ The catalyst was commercial aluminiumactivated TiCl₃; the cocatalyst was triethylaluminium.

Dodecene-1 concentration at the beginning of the polymerizations was in the range of $0.12-0.73 \text{ mol/dm}^3$. Polymerizations were carried out at 80 and 90°C. To study the effects of butene-1 concentrations, initial butene-1 concentration in the reactor was varied between 0.03 and 0.48 mol/dm³. Reaction conditions are given in Table I. Analytical methods have been described earlier.¹

RESULTS AND DISCUSSION

The ¹³C-NMR spectra of typical copolymer samples of ethylene/dodecene-1 and ethylene/butene-1/dodecene-1 appear in Figure 1. Spectra b and c indicate two different comonomer ratios. The reaction of dodecene-1 can be









Fig. 1. Typical ¹³C-NMR spectra of copolymers. Resonances indicated as ppm TMA. (a) Ethylene/dodecene-1 copolymer (experiment no. 2); (b) Ethylene/butene-1/dodecene-1 copolymer (experiment no. 9); (c) Ethylene/butene-1/dodecene-1 copolymer (experiment no. 11).

detected in the spectra from resonances typical for branches containing more than six carbon atoms at 38.2, 34.6, 32.2, 30.4, 27.3, 22.8, and 14.1 ppm TMS.² The DSC curves in Figure 2 show a melting range at about 100°C, the size of which increases with increasing dodecene-1 content. Table II gives the properties of prepared samples. Crystallinity has been reduced remarkably, when the comonomer content in the polymer has been increased. Crystallinities in Figure 3 are presented as a function of dodecene-

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			$\frac{\overline{M}_{w}}{\overline{M}_{n}}$	6.2	6.3	6.5	8.0	8.5	7.6	7.7	10	6.8	8.2	7.9	5.7	7.3	9.3
TABLE II Polymer Composition and Basic Properties			M_{ν} 10 ⁵ g/mol	4.0	3.6	1.8	1.4	1.5	1.2	1.3	2.3	1.2	1.3	1.5	1.1	0.9	1.2
			Density (g/dm³)	> 944	938	930	926	924	923	930	930	922	913	908	905	915	913
			Crystallinity (%)	67	60	59	48	46	44	48	54	40	38	32	30	39	35
			T _m (°C)	136	132	130	125	126	125	127	128	125	127	125	124	125	125
		Dodecene-1	concentration (wt %)	<3	< 3	2^{-5}	9.6	7.4	10	.^	<3	2.8	8.5	8.0	10	8.0	8.1
	mposition		Branches 100°C	< 0.2	< 0.2	0.1 - 0.3	0.9	0.7	1.0	< 0.1	< 0.2	0.3	0.8	0.7	1.0	1.5	0.8
	Polymer co	Riitana.1	concn (wt %)	1	1	0.9	1.6	4.6	5.0	3.4	3.0	5.4	7.0	6.2	6.0	5.3	6.7
			Ethyl branches/100°C	1	I	0.2	0.4	1.2	1.5	0.9	0.8	1.4	2.0	1.7	1.6	1.5	1.8
			Experiment no.	1	2	ę	4	5	9	7	œ	6	10	11	12	13	14

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Fig. 3. Effect of dodecene-1 content of copolymers on crystallinity. (\bullet) ethylene/butene-1/dodecene-1 copolymer (butene-1 content 5 wt %); (\blacksquare) ethylene/butene-1 copolymer (butene-1 content 5 wt %); (\blacktriangle) ethylene/dodecene-1 copolymer.

1 for samples polymerized at 90°C, and a clear reduction in crystallinity with increasing dodecene-1 content can be seen. When comonomer contents increased, the main melting peak temperature decreased to 124-125°C. Reduction of melting temperature according to density is shown in Figure 4. The lowest densities obtained were 905-910 g/dm³. The dependence of density on dodecene-1 content is shown in Figure 5. Crystallinity and density have a clear dependence on each other. According to the DSC curves, measured crystallinity decreased from 50 to 30% when density was lowered from 930 to less than 910 g/dm³.

Dodecene-1 produced the same effect as detected in terpolymerizations with decene-1. Dodecene-1 was more reactive when butene-1 was present in the reactor. Experiments were carried out at 80 and 90°C, but the effect of the polymerization temperature was small. It is also clearly seen that dodecene-1 content in the polymer remained at a low level when butene-1 was not present in the reactor. In Figure 6, the effect of initial butene-1 concentration in the reactor to the polymers butene-1 and dodecene-1 contents is shown. Initial dodecene-1 concentration was kept constant at 0.48 mol/dm³. Dodecene-1 content in the polymer increased from less than 3



Fig. 4. Dependence of copolymer melting temperature on dodecene-1 content of copolymer. Symbols are explained in Figure 3.





wt % to 8-10 wt %, when initial butene-1 concentration changed from zero to over 0.2 mol/dm³. When butene-1 concentration in the reactor was further increased, dodecene-1 concentration stabilized at a constant level. Butene-1 content in the polymer continued to increase in the same way as with ethylene/butene-1 copolymers.

In polymerizations with high butene-1 concentrations in the reactor, the increment in the amorphous polymer was substantial. This was evident in the crystallinities measured by DSC and in the formation of soluble polymer, which caused reactor fouling. The composition of both soluble and insoluble polymer was analyzed from ethylene/butene-1/dodecene-1 terpolymers. The suspension taken out of the reactor after the reaction period was divided into two parts. One part was treated as earlier described.¹ The other part was filtrated without the addition of isopropanol, so that the heptane soluble polymer was obtained in the solution. After filtration it was precipitated by isopropanol. The properties of the fractions are given in Table III.



Fig. 6. Effect of initial butene-1 concentration in the reactor on comonomer content measured from the terpolymers. Polymerization temperature was 90°C, and dodecene-1 concentration in the reactor was 0.48 mol/dm³: (\bigcirc) butene-1 content; (\bigcirc) dodecene-1 content.

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	Whole sample	Heptane- insoluble fraction	Heptane- soluble fraction		
Weight (g)	74	60	14		
Density (g/dm ³)	905	920			
Crystallinity (%)	30	43	< 5		
Melting temp (°C)	124	124	_		
Butene-1 content (wt %)	6	4.5	13		
Dodecene-1 content (wt %)	10	6.5	25		

TABLE III Properties of Sample 14: Its Heptane-Soluble and -Insoluble Fractions

About 40% of butene-1 was in the heptane soluble part, which was 19% of the whole sample. In this fraction was located 47% of dodecene-1. Because the soluble fraction had almost completely amorphous structure, it caused a reduction of crystallinity in the whole sample. Density was substantially lower, when the amorphous part was in the sample. The DSC curve of the heptane insoluble part was similar to that of the entire sample, having also the additional melting range at about 100°C.

To prevent the formation of soluble, amorphous polymer, while attempting to get high contents of dodecene-1 in the polymer, it is advantageous to keep the initial butene-1 concentration at about $0.05-0.15 \text{ mol/dm}^3$, while dodecene-1 concentration is 0.5 mol/dm^3 , and ethylene concentration around 0.5 mol/dm^3 .

Further information has been obtained by using pyrolysis hydrogenation gas chromatography to study the structure of the samples prepared in this study. These results have been reported earlier.^{3,4} The literature discusses application of this method to polyethylene analysis.⁵⁻⁸ In PHGC analysis, abnormally high concentrations of fractions caused by the long α -olefin were detected, when ¹³C-NMR and IR analysis were used as references. This indicates that *n* is small in the structure shown in Figure 7, where chain scission in pyrolysis evidently occurs at point A. The PHGC results indicate that branches caused by butene-1 are probably located near branches caused by the long α -olefin.

This result may explain why the reaction of long α -olefins increased significantly when a small amount of butene-1 was added at the same time.



Fig. 7. Structure of terpolymer, and probable chain scission point in PHGC.

We can assume that butene-1 causes a sterically less-hindered structure in the growing chain, allowing long α -olefins to react. Besides, there will be butene-1 comonomer branches along the polymer chain, the amount depending on butene-1 concentration.

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